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Effect of Running-In (Load and Speed) on Surface Characteristics of Honed Gears

Dinesh Mallipeddi\textsuperscript{a} \& Mats Norell\textsuperscript{b}, Mario Sosa\textsuperscript{b}, and Lars Nyborg\textsuperscript{a}

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**ABSTRACT**

The initial running-in cycles alter the surface integrity characteristics and influence gear performance. This article shows how the surface characteristics of honed spur gears evolved due to the combined effect of running-in load (0.9 or 1.7 GPa) and speed (0.5 or 8.3 m/s) in Forschungsstelle für Zahnräder und Getriebebau tests. Running-in affected the surface layers to a depth of 5 µm. High running-in load promoted plastic deformation of asperities, created microstructural changes associated with surface cracks, and relaxed residual stresses. It also enhanced the amount of phosphorous from extreme pressure (EP) additives at the surface. The surface contact fatigue failure—that is, micropitting—was promoted by running-in speed rather than load.

**ARTICLE HISTORY**

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**KEYWORDS**

Running-in; surface integrity characteristics; honed spur gears

**Introduction**

The demand for gear drive performance is high in terms of durability, efficiency, and load-carrying capacity. The surface integrity characteristics such as topography, residual stresses, and material structure generated by hard finishing determine the performance of transmission gears. For example, Tönshoff and Marzenel (1) produced three sets of gears with different roughness and residual stress states by varying gear honing parameters. Pitting tests on these gears showed different lifetimes. In addition to manufacturing, the operating conditions during initial cycles—that is, running-in—further govern the performance of gears by altering the initial surface characteristics.

In general, the torque is initially carried by a small fraction of asperities. This leads to stress concentrations over a narrow region and, depending on the operating conditions used, numerous micropits can initiate at the asperity level. Investigations by Ariura, et al. (2) revealed that micropitting, or grey staining, causes profile degradation. Therefore, further smoothening of asperities is required for better formability but preferably in a way that avoids the formation of micropits. The real contact area of gears manufactured by different methods such as green shaving, grinding, and honing increases with running-in (Sjöberg, et al. (3)). Andersson (4) observed that running-in decreased the roughness of both hobbed and shaved gears. However, the decrease was higher for hobbed gears due to their rougher initial surface.

Running-in changes the surface topography of virgin surfaces by plastic deformation and wear (Ismail, et al. (5)). The surface modification created by running-in attains a steady-state condition between contact pressure, surface roughness, and interface layer and also helps to establish an effective lubricating film at the interface (Hsu, et al. (6)). Kragelsky, et al. (7) reported that the effectiveness of running-in in deforming the asperities is influenced by factors such as load, speed, physical properties of the material, and lubrication medium. Sjöberg, et al. (8) found that the efficiency of ground gears was enhanced by high running-in load, but these gears also had more micropits and subsurface cracks, as shown by Mallipeddi, et al. (9).

The importance of the running-in process is clearly emphasized in the literature. However, most studies focus on specific phenomena; for example, roughness, residual stresses, or friction. Though these factors are affected as a consequence of deformation mechanisms, changes in material structure, and surface chemistry, no study of the combined changes induced by different running-in parameters was found. As a forward step, this research work focuses on the effect of running-in load and speed combinations on the surface integrity characteristics of honed gears. Different techniques were used to examine topography, residual stresses, microstructure development, and evolution of surface chemistry. This article will show how some changes are promoted primarily by load and others by speed.

**Material and experiment procedure**

**Materials, manufacturing, and gears**

The tested spur gears of alloy 16MnCr5 were case hardened, tempered, and hard finished first by generating grinding and then by a honing process. The case depth was about 1 mm

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Review led by R. Errichello.

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Table 1. Geometrical parameters of tested gears.

<table>
<thead>
<tr>
<th>Parameter</th>
<th>Gear</th>
<th>Pinion</th>
</tr>
</thead>
<tbody>
<tr>
<td>Number of teeth</td>
<td>24</td>
<td>16</td>
</tr>
<tr>
<td>Pressure angle (°)</td>
<td>20</td>
<td></td>
</tr>
<tr>
<td>Center distance (mm)</td>
<td>91.5</td>
<td></td>
</tr>
<tr>
<td>Module (mm)</td>
<td>4.5</td>
<td></td>
</tr>
<tr>
<td>Face width (mm)</td>
<td>14</td>
<td></td>
</tr>
<tr>
<td>Pitch diameter (mm)</td>
<td>109.8</td>
<td>73.2</td>
</tr>
<tr>
<td>Tip diameter (mm)</td>
<td>118.4</td>
<td>82.5</td>
</tr>
<tr>
<td>Tip relief starting Diameter (mm)</td>
<td>115.9</td>
<td>80.3</td>
</tr>
<tr>
<td>Tip relief (µm)</td>
<td>20</td>
<td></td>
</tr>
<tr>
<td>Profile shift</td>
<td>0.171</td>
<td>0.182</td>
</tr>
</tbody>
</table>

Table 2. Detailed test matrix with designations.

<table>
<thead>
<tr>
<th>Test designation</th>
<th>Load stage (torque, Nm)</th>
<th>Speed (m/s)</th>
</tr>
</thead>
<tbody>
<tr>
<td>R15-LS</td>
<td>x</td>
<td>x</td>
</tr>
<tr>
<td>R15-HS</td>
<td>x</td>
<td>x</td>
</tr>
<tr>
<td>R19-LS</td>
<td>x</td>
<td>x</td>
</tr>
<tr>
<td>R19-HS</td>
<td>x</td>
<td>x</td>
</tr>
</tbody>
</table>

with a surface hardness of about 750 HV. The gear geometry was similar to that of standard Forschungsstelle fur Zahnräder und Getriebebau (FZG) C-Pt gears (Höhn, et al. (10)) with a few modifications; the tip reliefs $c_{a1}$ and $c_{a2}$ were both 20 µm and the helix angle modifications $H/f1$ and $H/f2$ were about 6 µm (equal to a helix angle of 0.0327°). The nominal geometrical parameters of the tested gears are presented in Table 1.

**Test rig and running-in testing**

The running-in tests were performed using an FZG back-to-back test rig (Sjöberg, et al. (8)). It is important to note that a new pair of gears was used in each test and all analyses were performed on the driven gear wheel.

The running-in cycles were 20,880 revolutions of the wheel, which is standard for an Forschungsvereinigung Antriebstechnik efficiency test procedure (Group F-R (11)). The testing combined two different torques (94 and 302 Nm corresponding to Hertzian pressures at pitches of 0.9 and 1.7 GPa, respectively) and at low speed (LS) and high speed (HS; 0.5 and 8.3 m/s corresponding to 87 and 1,444 rpm). The torques are referred as load stage 5 (LS5) and load stage 9 (LS9) in FZG testing. The gears were fully dipped in a synthetic poly-alpha olefin lubricant with a density of 837 kg/m³ and nominal viscosity of 11.8 cSt at 100°C. The lubricant temperature during testing was maintained at 90°C. A detailed test matrix with the designation of individual tests is presented in Table 2, where, for example, R15-LS designates running-in at load stage 5 at low speed.

**Characterization methods**

Gear teeth for all analyses were mechanically cut from the wheel using an abrasive disc for metallographic sample preparation.

The surface roughness was characterized by a Form Talysurf PGI 800 (Taylor Hobson) profilometer using 0.15-µm stylus tip. Starting from the center of the face width of a gear tooth, three profiles were taken 0.1 mm apart. Two randomly selected teeth were measured for every condition. A measurement length of 6 mm was used for each profile measurement. To separate the form from waviness and roughness, a polynomial of order six was used. Thereafter, a Gaussian filter with a cutoff length of 0.8 mm was employed to separate the waviness from roughness.

The measured roughness was used to calculate the specific film thickness ($\lambda$), which is the ratio between the minimum film thickness and the composite roughness as defined by Gohar (12):

$$\lambda = \frac{h_{\min}}{\sqrt{R_{q1}^2 + R_{q2}^2}},$$

where $R_{q1}$ and $R_{q2}$ are the root mean square roughness parameters of the pinion and gear and $h_{\min}$ is the minimum film thickness as detailed in Andersson, et al. (13).

The surface topography and microstructure of the gear teeth were characterized using a scanning electron microscope operated at 5 kV. Secondary electrons were used for imaging both surface topography and microstructure.

Residual stresses were measured by means of X-ray diffraction using a Cr-Kα source. The lattice deformation for the [211] α-Fe peak was measured and stress values were determined using a standard sin² (ψ) technique (Fitzpatrick, et al. (14)) with five equi-sin² (ψ) tilts from −45°/+45°. Stepwise etching electrochemically using 3 M NaCl was used for depth profiling. For retained austenite measurements, the diffraction planes {200} and {220} were analyzed. The X-rays were irradiated through a circular 3-mm collimator and exposed for about 40 s with an inclination of +45° and scanning ±85° with 15 steps.

The tribolayers were analyzed by X-ray photoelectron spectroscopy (XPS) with a monochromatic Al Kα (1,486.6 eV) source on an analysis area ~0.8 mm in diameter. For depth profiles, Ar⁺ etching was used with rates as calibrated on Ta₂O₅. The gear teeth were cleaned in an ultrasonic bath for 5 min each with xylene and ethanol.

**Results and discussion**

**Surface roughness and topography**

The surface roughness parameters of an as-honed gear and after different running-in conditions are presented in Table 3. The roughness changes were characterized based on the amplitude parameters $R_a$, $R_q$, and Abbott-Firestone parameter $R_{pk}$. $R_q$ is used to calculate the specific film thickness. Overall, slightly smoother surfaces were measured after running-in.

However, the differences in surface deformation between running-in conditions observed by scanning electron microscopy were not clearly reflected in the roughness parameters. This can be attributed to the inherent limitation of profilometry to only measure roughness normal to the contact between the stylus and the surface on a single profile. Hence, detailed topographic examination by scanning electron microscopy is vital to understand the deformation and evolution of gear surfaces. In addition, an added
advantage is the possibility of mapping the gear surface along the axial direction.

The surface topography of as-honed gears is shown in Fig. 1. The surface lay orientation is different between regions of the gear flank. At pitch, the lay is perpendicular to the profile direction, like that of a ground gear. At both the tip and dedendum, the lay is oriented at an angle with respect to the pitch surface but with orientations in opposite directions. This is because the honing tool motion is cross-axed to the gear surface. Irrespective of orientation, the surface lay consists of adjacent peaks and valleys with irregular asperities. These surface microfeatures determine the contact area during the initial cycles.

The topography for different running-in conditions at the dedendum close to the end of the active profile is shown in Fig. 2. The micrographs reveal that surface asperities are plastically deformed for all test conditions. The deformation was influenced more by load than speed so that a higher load resulted in greater deformation. Moreover, micropits occurred after running-in for all test conditions. This is because the honing tool motion is cross-axed to the gear surface. Irrespective of orientation, the surface lay consists of adjacent peaks and valleys with irregular asperities. These surface microfeatures determine the contact area during the initial cycles.

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Irrespective of load, the calculated $\lambda$ values were higher for high speed conditions RI5-HS and RI9-HS compared to those at low speed conditions RI5-LS and RI9-LS; see Table 4. This indicates a better separation between mating gear pairs when run at high speed. Still, a smoother surface was obtained for high load and low speed (RI9-LS), whereas a large number of micropits was observed for high load and high speed (RI9-HS). Li and Kahraman (17) also reported that running-in with high contact pressure and lower rolling velocity reduced the number of micropits for case-hardened ground twin disks. Potential reasons for this behavior were gradual surface polishing and the formation of a tribofilm with lower boundary friction that reduces the surface shear. It is reasonable that a smoother surface provides lower friction. However, analyses of the tribofilm did not indicate a major difference in tribofilm formation with load between RI9-LS and RI9-HS or RI5-LS and RI5-HS in this study.

| Table 3. Surface parameters for different conditions. |

<table>
<thead>
<tr>
<th></th>
<th>As-honed mean value</th>
<th>RI5-LS mean value</th>
<th>RI5-HS mean value</th>
<th>RI9-LS mean value</th>
<th>RI9-HS mean value</th>
</tr>
</thead>
<tbody>
<tr>
<td>Tooth 1</td>
<td>Tooth 2</td>
<td>Tooth 1</td>
<td>Tooth 2</td>
<td>Tooth 1</td>
<td>Tooth 2</td>
</tr>
<tr>
<td>$R_a$</td>
<td>0.1825</td>
<td>0.1751</td>
<td>0.1539</td>
<td>0.1565</td>
<td>0.1494</td>
</tr>
<tr>
<td>$R_s$</td>
<td>1.5896</td>
<td>1.327</td>
<td>1.2828</td>
<td>1.1755</td>
<td>1.2554</td>
</tr>
<tr>
<td>$R_{pk}$</td>
<td>0.2336</td>
<td>0.1806</td>
<td>0.0991</td>
<td>0.1218</td>
<td>0.1022</td>
</tr>
<tr>
<td>$R_q$</td>
<td>0.2387</td>
<td>0.2268</td>
<td>0.2012</td>
<td>0.2040</td>
<td>0.1968</td>
</tr>
</tbody>
</table>

Irrespective of load, the calculated $\lambda$ values were higher for high speed conditions RI5-HS and RI9-HS compared to those at low speed conditions RI5-LS and RI9-LS; see Table 4. This indicates a better separation between mating gear pairs when run at high speed. Still, a smoother surface was obtained for high load and low speed (RI9-LS), whereas a large number of micropits was observed for high load and high speed (RI9-HS). Li and Kahraman (17) also reported that running-in with high contact pressure and lower rolling velocity reduced the number of micropits for case-hardened ground twin disks. Potential reasons for this behavior were gradual surface polishing and the formation of a tribofilm with lower boundary friction that reduces the surface shear. It is reasonable that a smoother surface provides lower friction. However, analyses of the tribofilm did not indicate a major difference in tribofilm formation with load between RI9-LS and RI9-HS or RI5-LS and RI5-HS in this study.

The topography of the tip surfaces is shown in Fig. 4. It is evident from the micrographs that the deformation pattern is different from that of dedendum surfaces. For low load conditions, irrespective of speed, the deformation of asperities is limited, and the topography resembles that of the as-honed condition. On the other hand, asperities were deformed plastically and micropits appeared for high load conditions. Though it is hard to quantify, more micropits were present in combination with high speed.

Speed promoted micropit formation at the dedendum where there is negative sliding (sliding and rolling moves in
the opposite direction), and load influenced micropitting at the tip surface where sliding is positive (sliding and rolling moves in the same direction).

**Residual stresses**

Residual stresses are an important factor in surface integrity that highly influences the fatigue life of gears. Hence, stress levels obtained after hard finishing and their evolution

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**Table 4.** Specific film thickness values for different running-in conditions.

<table>
<thead>
<tr>
<th></th>
<th>RI5-LS</th>
<th>RI5-HS</th>
<th>RI9-LS</th>
<th>RI9-HS</th>
</tr>
</thead>
<tbody>
<tr>
<td>$\lambda$</td>
<td>0.17</td>
<td>1.25</td>
<td>0.17</td>
<td>1.10</td>
</tr>
</tbody>
</table>

*The specific film thickness is calculated based on the composite roughness values of tooth 1.*
during initial cycles were closely monitored. Figure 5 summarizes the residual stresses on the dedendum surface for all tests with average values from three different teeth, differing at most by ±20 MPa. For as-honed gears, the stresses were compressive in both directions and slightly higher in profile direction. After running-in, the stresses tend to decrease in both directions, more so in the profile direction. Tönshoff, et al. (18) explained that maximum load stresses act in the profile direction due to high frictional forces created by combined rolling and sliding. Consequently, severe local plastic deformation in the profile direction results in higher stress variations. For ground gears, Mallipeddi, et al. (9, 19) observed increasing compressive residual stresses in both directions after running-in, which relaxed during further testing, at least in the profile direction. It is important to note that grinding induced much lower stresses than honing. This shows that the evolution of stresses is also dependent on the initial stress state and its response to material yielding.

The lowest compressive stresses were observed for R19-LS. Overall, load had a great influence on stress relaxation, and for both loads the combination with low speed resulted in greater stress relaxation. Running-in also resulted in a stress gradient between axial positions. This is attributed to the helix angle modification that resulted in a nonuniform pressure distribution along the width of the flank.

Residual stress depth profiles are shown in Fig. 6. The case stress was around 200 MPa, equal in both directions. Honing increased the stresses in the outermost ~10 μm, whereas the relaxation induced by running-in was limited to a depth of ~5 μm. Similar results were reported by Tönshoff, et al. (18). In that study, honed gears made of the
same material were tested for pitting life. Running-in was performed with a torque of 132 Nm, intermediate between the torques used in this study. In addition to the nominal shear stress arising from the nominal geometry, mating asperities generated shear stresses close to the surface. Sosa (20) simulated the running-in process of ground gear surfaces and the predicted results depicted higher shear stresses confined to the outermost layers of \( \approx 7 \mu m \). Previously, the authors (Mallipeddi, et al. (21)) characterized the surface integrity of efficiency-tested ground, honed, and superfinished gears. The results showed that the testing relaxed compressive residual stresses only for ground and honed gears, whereas for smoother superfinished gears the residual stresses remained unaltered. Moreover, the relaxation for ground and honed gears was limited to a depth of \( \approx 5 \mu m \). This implies that the evolution of residual stresses is related to the variation in pressure distribution of mating asperities and their consequent deformation.

**Microstructure**

For as-honed gears, the microstructure of the case-hardened layer consists of plate martensite. The surface-retained austenite content was about 9% as determined by X-ray diffraction. After running-in, a distinct structural feature was observed near the tooth surface (to a depth of \( \approx 5 \mu m \)) along the profile but mostly at the dedendum. An example is shown in Fig. 7. The morphology consists of thin parallel bands oriented in different directions and sometimes intersecting.

Though it is hard to quantify, more bands were seen for RI9 tests and in particular for RI9-LS. In a previous study (Mallipeddi, et al. (9)), the authors observed these features in ground gears that were run-in under the same test conditions. The morphology of these bands appears similar to that of deformation-induced martensite in stainless steel (Das, et al. (22)). However, this is not likely to be the case in this study because the retained austenite content remains within 8–9% from as-honed to running-in test conditions.

Detailed cross section analyses also revealed the association of these features with surface cracks (Fig. 8), known to be connected to asperities/the surface lay (Mallipeddi, et al. (9)). This indicates that these features could be deformation bands formed due to cyclic plastic deformation of asperities, from which surface fatigue cracks initiated. Nevertheless, detailed transmission electron microscopy analyses would be required to identify the structure and determine its influence on the performance of gears.

**Surface chemistry**

The major role of a lubricant is to protect and minimize friction between gear pairs by avoiding metal-to-metal contact via establishing a tribofilm. Depending on the operating conditions and topographical changes, direct metal contact can be prevented by the formation of a reaction layer with extreme pressure (EP) additives in the lubricant.

The results of surface chemical analysis are presented as XPS depth profiles. The analysis area was about 0.8 mm in diameter and there was some roughness on the surface of the samples. Ion etching is influenced by the roughness and the profiles represent the average composition integrated over layers and particles with different compositions and thicknesses. The profiles (not presented) of as-honed gears showed the presence of an Fe oxide limited to a few nanometers, the alloying elements, and some C contamination. Characteristic elements that might originate from the cutting fluid were N (\( \approx 3 \) at%), Si (\( \approx 5 \) at% at the surface), and traces of Ca. Nitrogen was mostly present at the surface, whereas Si was present down to 5 nm. No P or S was found on the as-honed gear.
A tribofilm develops depending on the testing, and the most interesting elements to follow are P and S from EP additives. Phosphorous was observed after running-in as shown in Fig. 9. Higher concentrations (~7 at%) and thicker layers (~11 nm) were recorded for high running-in load, independent of speed. This is in conjunction with the evolution of surface topography where severe deformation was seen for higher running-in load. At low load, significantly less P was recorded in similar concentrations for both speeds. The other main EP additive constituent, S, was not found in any test condition. Overall, the load clearly influenced the tribofilm formation, whereas it appears unaffected by speed.

Conclusions

Running-in certainly affected the surface characteristics in a layer limited to <5 μm. The surface asperities were deformed through plastic deformation and the severity increased with increasing load. In addition, the formation of thin parallel bands in the microstructure, as well as relaxation of the residual stresses, were also promoted by the load. Surface cracks were observed to be associated with structural changes. A higher amount of P from EP additives was observed for higher running-in load conditions.

Higher running-in load significantly improved the smoothness of the surface. However, it simultaneously affected the surface integrity by inducing microstructural changes that have a detrimental effect on the life of gears. On the contrary, micropitting readily occurring during running-in was enhanced more by speed than load. Hence, the micropitting mechanism was speed dependent. Overall, this study shows how a running-in process alters the hard finishing surface characteristics and presents the conditions that determine the performance of gears.

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References


