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# Bottom-Up Growth of Monolayer Honeycomb SiC

C. M. Polley,<sup>1,2,\*</sup> H. Fedderwitz,<sup>1</sup> T. Balasubramanian<sup>1</sup>, A. A. Zakharov,<sup>1</sup> R. Yakimova<sup>1,3</sup>, O. Bäcke,<sup>2</sup> J. Ekman<sup>1,2</sup>, S. P. Dash,<sup>2</sup> S. Kubatkin,<sup>2</sup> and S. Lara-Avila<sup>2,4</sup>

<sup>1</sup>MAX IV Laboratory, Lund University, Fotongatan 2, 22484 Lund, Sweden

<sup>2</sup>Department of Microtechnology and Nanoscience, Chalmers University of Technology, 412 96, Gothenburg, Sweden

<sup>3</sup>Department of Physics, Chemistry and Biology (IFM), Linköping University, Linköping, 581 83 Sweden

<sup>4</sup>National Physical Laboratory, Hampton Road, Teddington TW11 0LW, United Kingdom



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The long theorized two-dimensional allotrope of SiC has remained elusive amid the exploration of graphenelike honeycomb structured monolayers. It is anticipated to possess a large direct band gap (2.5 eV), ambient stability, and chemical versatility. While  $sp^2$  bonding between silicon and carbon is energetically favorable, only disordered nanoflakes have been reported to date. Here we demonstrate large-area, bottom-up synthesis of monocrystalline, epitaxial monolayer honeycomb SiC atop ultrathin transition metal carbide films on SiC substrates. We find the 2D phase of SiC to be almost planar and stable at high temperatures, up to 1200 °C in vacuum. Interactions between the 2D-SiC and the transition metal carbide surface result in a Dirac-like feature in the electronic band structure, which in the case of a TaC substrate is strongly spin-split. Our findings represent the first step towards routine and tailored synthesis of 2D-SiC monolayers, and this novel heteroepitaxial system may find diverse applications ranging from photovoltaics to topological superconductivity.

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Chemical bonding between C and Si may exhibit competition between the  $sp^3$  hybridization preferred by silicon and the  $sp^2$  hybridization preferred by carbon [1]. Nonetheless, first-principles calculations predict that two-dimensional,  $sp^2$  bonded SiC (2D-SiC) should be stable, with the planar, stoichiometric structure the most energetically favorable conformation [2–5]. Despite considerable interest in its properties and several theoretical studies [11–14], experimental synthesis of 2D-SiC has proven difficult [15–17]. Central to this challenge is that, unlike graphitic carbon, bulk SiC does not exist in a layered form and hence does not lend itself to top-down exfoliation techniques.

Here we demonstrate a bottom-up approach to the synthesis of 2D-SiC, by preparing ultrathin carbide films ( $t < 3$  nm) on SiC substrates followed by annealing to high temperature ( $T > 1700$  °C), similar to the process used for epitaxial growth of graphene on SiC. Underlying this approach is our ability to produce high-quality ultrathin epitaxial films of transition metal carbides on the silicon terminated face of 4H-SiC. Carbides are well known for their extreme thermal, chemical, and mechanical robustness; and among the transition metal carbides, TaC and

NbC in particular have received renewed attention owing to their bulk superconductivity ( $T_C \approx 11$  K) and topologically nontrivial electronic band structure [6–8]. Both TaC and NbC can grow epitaxially on 4H-SiC, as they have a rocksalt crystal structure with a threefold symmetric (111) plane that is nearly lattice matched to 4H-SiC (0001) (2.3% and 2.7% lattice mismatch, respectively). Huder *et al.* reported a bottom-up solid state reaction in which films of Ta on a SiC substrate were transformed into graphene-terminated TaC by thermal annealing, with carbon being sourced from the decomposing SiC [9,10]. While the simplicity of this approach is appealing, the resulting TaC film is polycrystalline with grain sizes of  $\approx 200$  nm<sup>2</sup>. We have developed an alternative process involving sputter deposition of approximately ten monolayers of transition metal carbide followed by annealing to 1700 °C in an Ar atmosphere. Directly depositing a carbide rather than the base metal ensures a stoichiometric film and recrystallization upon annealing. Remarkably, we observe that while the annealing step indeed structurally orders the carbide, in addition it leaves the surface terminated with a monolayer of honeycomb SiC.

**Results.**—The samples studied are schematically depicted in Fig. 1(a). They consist of 3 nm thick films of TaC or NbC sputter deposited onto Si-terminated 4H-SiC(0001) substrates, with a lateral extent of approximately 5 mm  $\times$  5 mm. The deposition was followed by a 1700 °C anneal for 10 min in an Ar atmosphere. Here we mainly focus on TaC, but later will show that a similar result is obtained from NbC. The TaC/SiC interface is abrupt and the films are epitaxial, as

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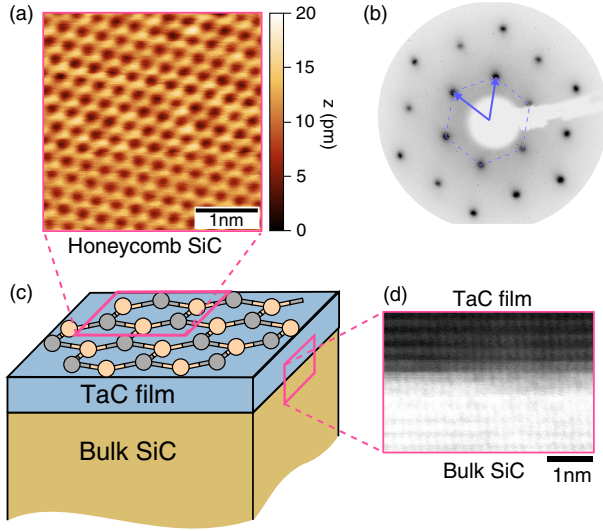


FIG. 1. (a) Topographic STM image of the monolayer honeycomb SiC layer terminating a TaC(111) surface ( $V_{\text{bias}} = -1.2$  V,  $I = 3$  nA). (b) Representative LEED image ( $E = 145$  eV);  $2.32 \text{ \AA}^{-1}$  unit cell reciprocal vectors are indicated. (c) Schematic depiction of the sample structure (d) Cross-sectional HAADF STEM image of the TaC/SiC interface.

revealed by cross-sectional high-angle annular dark-field scanning transmission electron microscopy (HAADF STEM). Cubic TaC has a rocksalt crystal structure with lattice parameter  $a = 4.456 \text{ \AA}$  [18], which presents  $\{111\}$  facets with an in-plane lattice parameter  $a = 3.15 \text{ \AA}$ , readily accommodated atop the SiC substrate with  $a = 3.08 \text{ \AA}$ . [19]. While hexagonal ( $\text{Ta}_2\text{C}$ ) or ternary ( $\text{Ta}_2\text{SiC}$ ) phases are in principle possible, there are several indications that the films as grown take the simple rocksalt structure. The HAADF STEM [Fig. 1(d)] shows cubic ordering, and the measured superconducting transition temperature of 9–10 K (Supplemental Material [20]) is consistent with cubic, stoichiometric TaC [24]. After transferring the samples into UHV and briefly annealing at  $1000^\circ\text{C}$  to remove surface oxidation, sharp hexagonal spots in LEED [Fig. 1(b)] and bands in ARPES (Fig. 2) become visible. Scanning tunneling microscopy (STM) [Fig. 1(a)] indicates a hexagonal surface lattice with  $a = (3.12 \pm 0.04) \text{ \AA}$ , uniformly covering the millimeter-scale sample (see also Supplemental Material). This surface is robust against longer or repeated heating cycles to similar temperatures, but at much higher temperatures will begin to transform into graphene (see Supplemental Material).

The most striking features in the ARPES measurements are spin-split surface states within a large surface-projected band gap of the TaC [Fig. 2(a)]. Photon energy scans [Fig. 2(b)] demonstrate the 2D character of these states, while constant-energy surfaces [Fig. 2(c)] confirm a  $1 \times 1$  hexagonal surface lattice (overlaid in red). The zone size ( $\bar{\Gamma}-\bar{K} = 1.34 \text{ \AA}^{-1}$ ) agrees with the lattice constant observed in STM. A high-resolution cut through  $\bar{K}$  [Fig. 2(d)] clearly

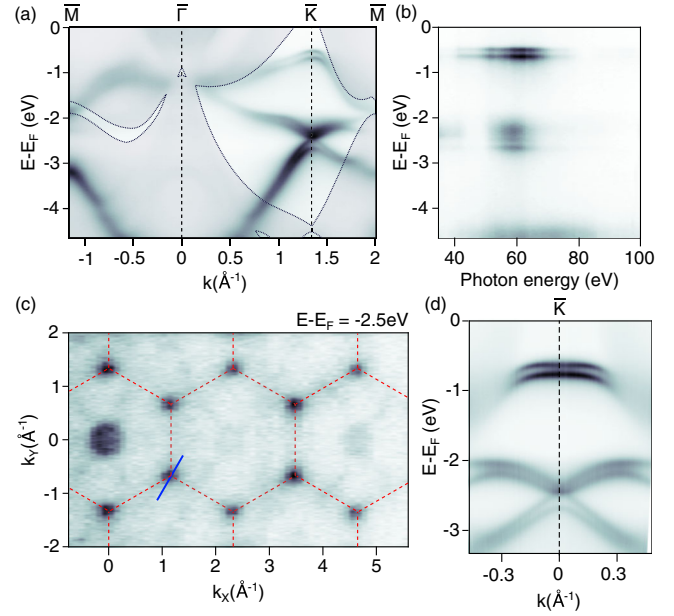


FIG. 2. (a) ARPES measurements ( $h\nu = 67$  eV,  $T = 80$  K) along the  $\bar{M}-\bar{\Gamma}-\bar{K}-\bar{M}$  path, overlaid with the surface-projected bulk band structure of TaC. (b)  $h\nu$  resolved cut at  $\bar{K}$  ( $h\nu = 35\text{--}100$  eV,  $T = 80$  K). (c) Constant-energy surface through the Dirac-like crossing of the surface state, with the surface Brillouin zone overlaid ( $h\nu = 180$  eV,  $T = 80$  K). (d) High-resolution ARPES measurement of the surface states at  $\bar{K}$ , direction indicated by the blue line in (c) ( $h\nu = 20$  eV,  $T = 19$  K).

illustrates the spin-orbit splitting of the bands, which reaches 125 meV for the upper bands and 220 meV for the lower set. All of these states are completely occupied, with the Fermi surface dominated by TaC bulk states spanning most of the surface Brillouin zone.

Based on evidence offered by LEED, STM, core levels (described below), and particularly DFT band structure calculations, we conclude that the ARPES measurements can only be explained by the presence of a monolayer of honeycomb SiC. The proposed structure is depicted in Fig. 3(a), and generates a calculated band structure that is in excellent agreement with experimental spectra [Figs. 3(b) and 3(c)]. The registry of the SiC monolayer has carbon atoms atop the substrate Ta atoms and silicon atoms in a fcc-hollow position. After relaxation along the  $c$  axis the SiC layer is weakly buckled ( $\Delta z = 16.8$  pm) and located  $2.3 \text{ \AA}$  above the surface layer of Ta atoms. In order for the DFT model to reproduce the degeneracy at  $\bar{K}$  the buckling must be reduced to 11.8 pm (see Supplemental Material). Unlike silicene (44 pm), the 2D-SiC here is nearly planar.

To interpret the band structure calculations it is helpful to draw comparisons with an isolated, planar honeycomb SiC monolayer [Fig. 3(d)]. Quite unlike its semimetallic relatives graphene or (planar) silicene, honeycomb SiC features a large band gap between the  $\pi$  and  $\pi^*$  bands. In Fig. 3(e) we plot the atom-, layer- and  $k$ -resolved band structure,

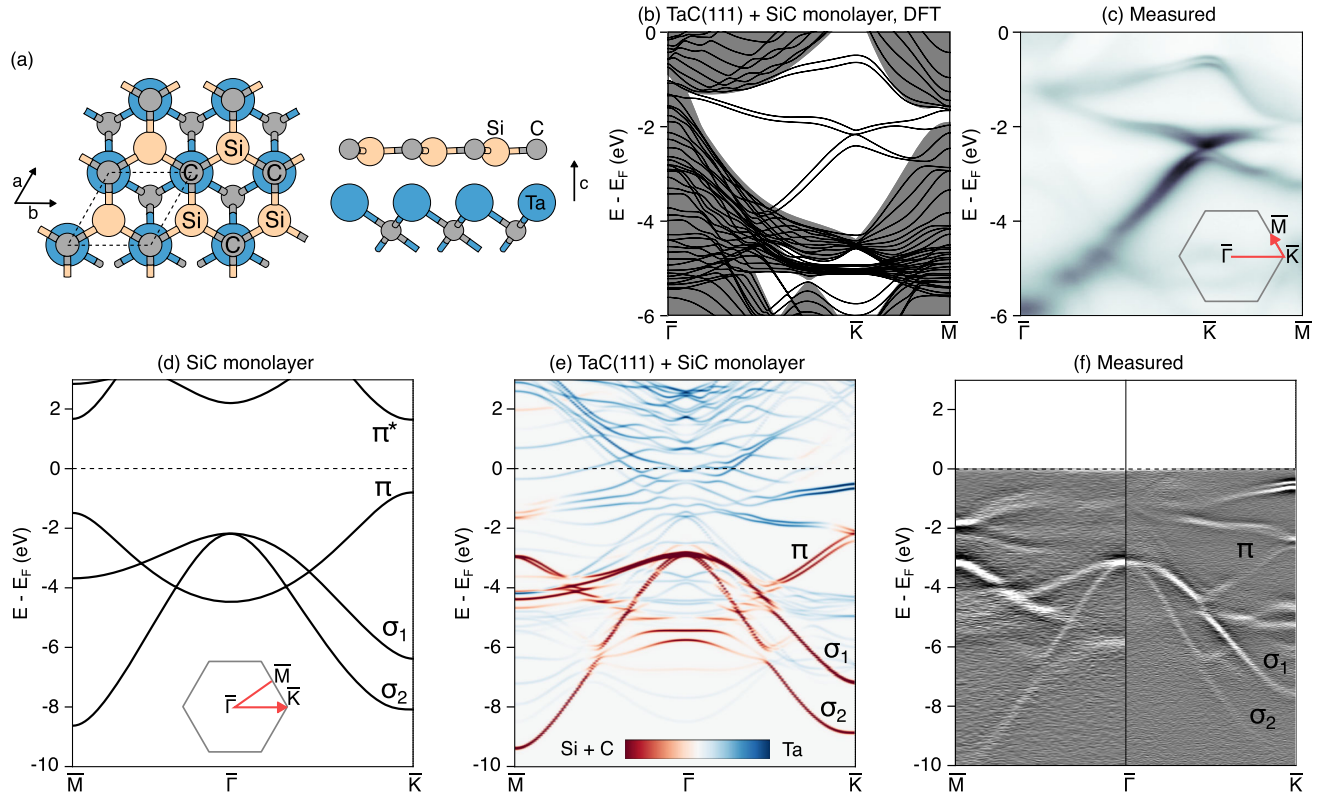


FIG. 3. (a) Schematic depiction of the honeycomb SiC structure. (b) Slab DFT band structure calculation for this structure along the  $\bar{\Gamma}-\bar{K}-\bar{M}$  path. The surface-projected TaC bulk band structure is indicated by the shaded grey regions. (c) ARPES spectrum at  $h\nu = 67$  eV,  $T = 80$  K. (d) DFT band structure of a planar, freestanding SiC monolayer along the  $\bar{M}-\bar{\Gamma}-\bar{K}$  path. (e) The same calculation for SiC atop TaC(111) as shown in (b), but sampling only the topmost two atomic layers and resolved into atomic character. (f) Corresponding experimental spectra, presented as a second derivative image. The  $\bar{\Gamma}-\bar{M}$  slice is acquired from the second Brillouin zone with  $h\nu = 138$  eV while the  $\bar{\Gamma}-\bar{K}$  slice is with  $h\nu = 44$  eV, both at  $T = 80$  K.

obtained by projecting the calculated wave functions onto the topmost two atomic layers of the slab (Si, C, and Ta atoms). This suppresses the bulklike TaC slab bands and thereby mimics the surface sensitivity of an ARPES measurement. We additionally use a color scale to represent the atomic character of each state. Presented in this manner, the SiC  $\sigma_1$  and  $\sigma_2$  bands become obvious both in the calculations and in experimental spectra [Fig. 3(g)]. We can furthermore identify the valence band of the Dirac-like surface state as deriving from the SiC  $\pi$  band, while the conduction band and also the uppermost surface state have strong Ta  $5d$  orbital character. The hybridization of the  $\pi$  band, together with the large spin-orbit splitting, makes it clear that the SiC layer is strongly interacting with the TaC substrate.

Core level spectroscopy measurements [Figs. 4(a)–4(c)] are consistent with the proposed structure in Fig. 3(a). The only species observed on the surface are Ta, Si, and C, and none are trace impurities—when measured together at  $h\nu = 650$  eV the cross-section adjusted intensity ratio is Ta:Si:C = 0.4:0.25:1. The Ta  $4f$  core level spectrum [Fig. 4(a)] consists of bulk ( $B$ ) and surface ( $S$ ) doublets, consistent with previous studies of  $\{111\}$  facets of bulk transition metal carbides that uniformly find a metal

element rather than carbon termination [25]. While the position of our  $B$  doublet is in excellent agreement with stoichiometric bulk TaC [26], the bulk-surface doublet splitting is approximately 250 meV smaller than would be expected from a simple TaC(111) surface [25]. The Si  $2p$  spectrum [Fig. 4(b)] consists of a single doublet and derives from silicon atoms close to the surface. This photon energy (130 eV) could in principle be weakly sensitive to silicon at the buried interface with the bulk SiC substrate, but this is unlikely to be the origin of the Si peak given that it shows an abrupt shift in binding energy when the surface oxide is removed (Supplemental Material). The C  $1s$  spectrum [Fig. 4(c)] contains three components, of which the component  $HC$  alone is closely correlated with the presence of the honeycomb SiC band structure. The other two components exist both before the SiC termination is established and after it is eliminated (see Supplemental Material). One of these peaks ( $B$ ) can be assigned to bulk carbon in the TaC film, with a binding energy in good agreement with measurements from the (001) surface of a bulk TaC crystal [26]. Figure 4(d) suggests origins for each component.

**Discussion.**—While the overall agreement between the DFT calculations and ARPES measurements is excellent,



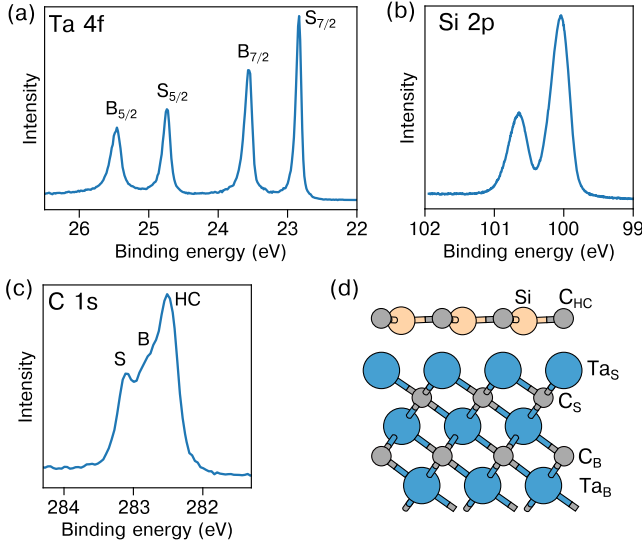


FIG. 4. High-resolution core level spectra. (a) Ta 4f at  $h\nu = 140$  eV ( $\Delta E = 10$  meV), (b) Si 2p at  $h\nu = 130$  eV ( $\Delta E = 5$  meV) and (c) C 1s at  $h\nu = 430$  eV ( $\Delta E = 80$  meV). All spectra acquired at  $T = 20$  K. (d) Suggested origin of the observed components.

there are two notable exceptions concerning the Dirac-like feature at  $\bar{K}$ . The first concerns the experimentally observed degeneracy—after full relaxation of the slab along the  $c$  axis, band calculations rather predict a sizable band gap of 200 meV. This can be resolved by reducing the buckling of the SiC monolayer from 16.8 pm to 11.8 pm, which is still larger than the  $< 5$  pm suggested by STM (see Supplemental Material). While it is interesting that the real system apparently assumes precisely the buckling that results in a band degeneracy at  $\bar{K}$ , the gap being a continuously tunable function of the buckling argues against the existence of a symmetry protection. The second exception concerns the binding energy of this feature, which requires a rigid shift of 280 meV to higher values to agree with experiments [Fig. 5(c)]. The Ta 5d derived surface state at lower binding energy does not require such a shift. We assume that these discrepancies can be resolved with a more sophisticated calculation method (e.g., DFT + U) or with a more complex structural model (e.g., including a buffer layer).

Close to the Dirac-like degeneracy at  $\bar{K}$ , it is possible to alternatively describe the dispersion using the  $k \cdot p$  model originally suggested for incorporating spin-orbit coupling effects into the graphene Dirac-cone dispersion [27,28]:

$$\epsilon = \mu\lambda_{\text{BR}} + \nu\sqrt{(\hbar vk)^2 + (\lambda_{\text{BR}} - \mu\lambda_I)^2} \quad (1)$$

where  $\nu = \pm 1$  indexes conduction and valence bands and  $\mu = \pm 1$  the spin-split bands,  $v$  is the band curvature while  $2\lambda_I$  and  $2\lambda_{\text{BR}}$  are, respectively, the intrinsic spin-orbit splitting and extrinsic Bychkov-Rashba splitting. Good agreement with the experimental data is found using

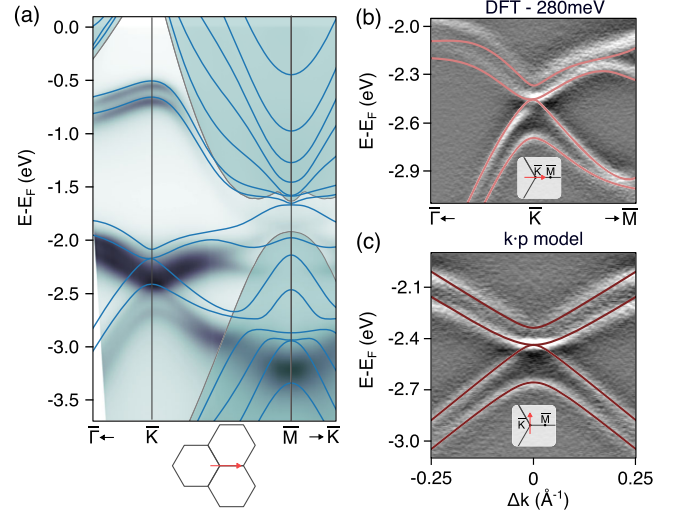


FIG. 5. (a) ARPES spectrum ( $h\nu = 23$  eV) overlaid with calculated band structure; shaded regions indicate surface-projected bulk bands of the TaC film. (b) Close-up of the Dirac-like crossing at  $\bar{K}$  ( $h\nu = 40$  eV), presented as a second derivative image and overlaid with the DFT band structure after applying an energy offset of 280 meV. (c) Further magnified spectrum of the Dirac-like crossing [ $h\nu = 20$  eV, from Fig. 2(d)], now overlaid with a fit to a  $k \cdot p$  model for graphene with spin-orbit coupling. All spectra acquired at  $T = 20$  K with  $\Delta E = 5$  meV.

$2\lambda_I = 60$  meV and  $2\lambda_{\text{BR}} = 160$  meV, which at  $k = 0$  (i.e.,  $\bar{K}$ ) corresponds to an absolute splitting in the valence bands of 220 meV. For reference, this is comparable to the largest reported band splittings observed in graphene following Au ( $\approx 100$  meV [29,30]) or Pb intercalation ( $\approx 200$  meV [31]). The sizable band splitting here derives from the heavy Ta atoms, readily confirmed by comparing with the same 2D-SiC termination that also forms on lighter niobium carbide films prepared in the same way (Fig. 6)

The atomistic details of how this honeycomb SiC layer forms remain to be clarified. If we consider the SiC substrate in isolation, graphitization by thermal decomposition is a complex but by now well-understood phenomenon [32]. At temperatures of  $\approx 1400^\circ\text{C}$  silicon atoms sublime from the surface while carbon, with a lower vapor pressure, does not. This leaves a carbon-rich layer that recrystallizes as graphene. In contrast, TaC is significantly more stable at high temperatures; to decompose it on a timescale of minutes would require extraordinary temperatures ( $> 3000^\circ\text{C}$  [33]) far from what has been reached here. Therefore we expect that the process is driven by the decomposition of the underlying SiC, leaving the TaC film intact. The most likely scenario is that the SiC termination forms during the postdeposition anneal of the TaC film to  $1700^\circ\text{C}$ , with Si and C diffusing through the TaC to accumulate on the surface. In this scenario, the sole function of the UHV annealing, at least for temperatures up to  $1000^\circ\text{C}$ , is to eliminate oxidation of the already-present SiC layer on

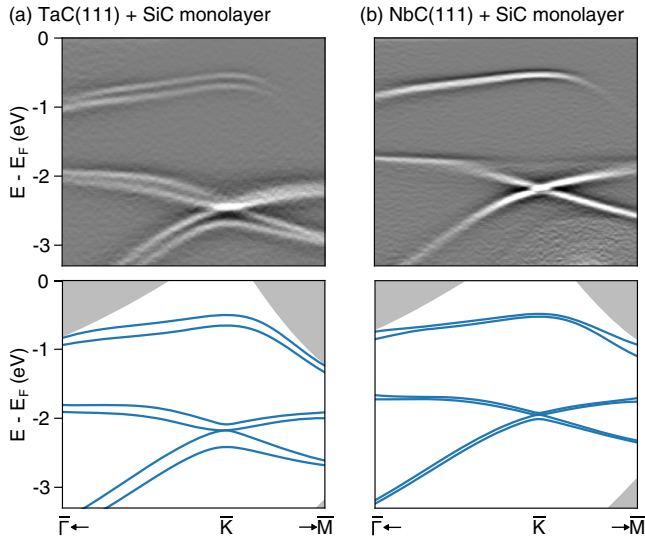


FIG. 6. (a) ARPES spectrum at  $h\nu = 35$  eV ( $T = 80$  K), presented as a second derivative image. Below is the calculated band structure. (b) The same for a SiC-terminated NbC film.

the surface. This is consistent with our observation that the surface condition is unaffected by longer or additional UHV anneals. However, this does not hold for much higher UHV annealing temperatures ( $>1200^\circ\text{C}$ ), for which the surface termination transforms into graphene (see Supplemental Material).

In conclusion, we have demonstrated a route to prepare large-area, high-quality SiC honeycomb monolayers through the annealing of thin films of transition metal carbides grown on  $4H$ -SiC(0001). We have shown this for TaC and NbC, but anticipate that the process is generic and may also occur with other suitably lattice-matched rocksalt monocarbides on SiC such as VC, TiC, or WC. Band structure calculations provide good overall agreement with ARPES measurements. Together they reveal that the  $\pi$  band of the SiC layer hybridizes with the underlying TaC surface states to yield a strongly spin-orbit split Dirac-like feature at the  $\bar{K}$  points. Beyond seeking a clearer understanding of the formation mechanism, in the future it will be interesting to explore possibilities of modifying this termination by intercalation or adsorption of other species. Such experiments may decouple the SiC layer from the substrate, alter the band splitting or even bring the Dirac-like feature towards the Fermi energy. The bottom-up fabrication technique shown here finally brings 2D SiC monolayers into the growing family of honeycomb structured monolayers.

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\*craig.polley@maxiv.lu.se

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