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Flores, E., Mozhzhukhina, N., Li, X. et al (2022). PRISMA: A Robust and Intuitive Tool for High-Throughput Processing of Chemical Spectra. CHEMISTRYMETHODS, 2(10).
<http://dx.doi.org/10.1002/cmt.202100094>

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PRISMA: A Robust and Intuitive Tool for High-Throughput Processing of Chemical Spectra**

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Here we introduce PRISMA, an open-source tool to rapidly analyse large numbers of spectra into meaningful spectroscopic trends. PRISMA follows a human-in-the-loop workflow, where the user interacts with an intuitive graphical user interface to control multiple steps in the spectrum analysis process: trimming, baseline correction, and peak fitting. The tool outputs the results in an easy-to-read csv format within seconds. We describe the functionalities implemented in PRISMA and test its capabilities with three experimental cases relevant to the study

of Electrochemical energy storage and conversion devices: a temperature-dependent Raman measurement of phase transitions, a linear Raman mapping of a graphite composite electrode, and an operando X-ray diffraction experiment of lithium nickel oxide composite battery electrode. The case studies demonstrate the robustness of the app and its ability to unearth insightful trends in peak parameters, which are easier to represent, interpret and further analyse with more advanced techniques.

Introduction

The rapid development of advanced tools for characterizing electrochemical energy storage and conversion devices (EESCDs) are popularizing high-throughput (HT) experiments. In HT experiments, a large number of measurements are acquired from different locations on a spatially-heterogeneous system (i.e., mapping) or from a system perturbed by temperature, electric potential, time, pressure, etc., to monitor its state, evolution, and degradation as a function of the perturbation variable. In particular, in situ and operando characterizations are increasingly used because the complex nature of the electrochemical environments requires studying functional materials within the working environment and while they operate. Typically, these experiments involve building a functional device (e.g., an electrolyser, a Li-ion battery, etc.) into a special cell that enables operating the device and characterizing its components through special windows transparent to the probing radiation.

A significant portion of the portfolio of techniques applied to study EESCDs^[1–5] output a one-dimensional array of (photon/electron) counts whose indexes correspond to a scanning variable, for example, energies in absorption spectroscopies, wavenumbers in vibrational spectroscopies, diffraction angles in diffraction-based techniques. Henceforth we use the term *pattern* to generalize any dataset sharing the aforementioned array data structure.

HT experiments can generate hundreds or even thousands of patterns as a function of a perturbation variable (e.g. time, temperature, etc.), which are often modulated by instrumental, measurement, and physical factors that introduce undesirable data features, for example, high noise, outliers, and unstable baselines. Conventionally, every spectrum is manually processed with basic techniques using commercial software (e.g., smoothing for noise reduction and outlier removal, polynomial baseline fitting to anchored points in the patterns), to then be represented into cascade or contour plots for drawing conclusions. When the experiment outputs several patterns, individual peak fitting is also useful to identify trends in peak parameters and deconvolute signals. However, as the number of patterns per experiment grows, a manual approach becomes not only cumbersome but also susceptible to arbitrary user decisions that can bias or negatively affect the reproducibility of the analysis. The automation of the whole analysis pipeline becomes necessary, but it is often difficult to implement in commercial software. Hence, many research groups resort to developing in-house and custom-made code that is rarely published, and even when it is, the code appears inaccessible to researchers with no programming experience. Ideally, any such tool should fulfil two main goals. The first is to assemble a library of spectrum-processing functions into an open codebase that can be operated, updated and populated by a broad community. The second goal is to make these tools accessible to researchers with no programming experience, via an intuitive graphical user interface (GUI). While there are notable tools

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[**] A previous version of this manuscript has been deposited on a preprint server (<https://doi.org/10.26434/chemrxiv-2021-7qs3m-v2>).

Supporting information for this article is available on the WWW under <https://doi.org/10.1002/cmt.202100094>

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available for processing spectra,^[6–9] these either lack GUIs, or they implement components built on proprietary dependencies, or do not support high-throughput analysis, or the code is compiled hence restricted from community development.

Here, we present PRISMA, an app based on Python for reproducible, high-throughput processing of multiple patterns in an automatable way. A prototypic version of the code was already demonstrated to be critical in the elucidation of dynamic phenomena on Li-ion battery electrodes^[10–12] and interfaces.^[13] We now build from previous versions to produce an externalizable app addressing the scientific need for open, reliable, user-friendly and fast analysis tools to extract meaningful patterns from (time, voltage, temperature, spatially)-dependent spectroscopic experiments. In the following sections we provide an overview of the code implementation initially supporting three main processing functionalities, designed as modular building blocks to facilitate future implementations of novel analysis techniques. In addition, we describe the general analysis workflow and present three case studies testing the robustness of the PRISMA app and demonstrating its role in analysing phase transitions, chemical maps and structural changes in temperature-, spatially- and time-dependent experiments, respectively. Finally, we discuss the value of PRISMA as a research tool and directions for future development.

Code Implementation and Use

PRISMA is designed with the vision of the BIG-MAP project, which aims to accelerate the discovery of new battery materials.^[14,15] The development of interoperable, community-wide research tools is essential within this vision.^[16] Accordingly, the design of the app is guided by the goals of facilitating the implementation, deployment and interoperability of spectrum-processing tools. Python and its numerical computing library Numpy are highly popular for scientific programming,^[17] and so they are obvious choices for implementing spectrum processing tools. In addition, we provide a GUI built with the ipywidgets module and rendered with the *voilà* package. For developers, the widget package enables building apps capable of running on the web browser using Jupyter Lab.^[18–20] At the same time, end-users (e.g., researchers) interact directly with clickable components in the app and so they are not burdened with implementation details. Two key design choices are made to support code interoperability. First, patterns and its processing tools are decorated with metadata identifiers such that they can be integrated into, for example, ontology^[21] and provenance management frameworks.^[22] Second, the GUI components are independent from all core functionalities. In this way, the spectrum processing functions (e.g., the baseline subtraction algorithm) can be operated not only by the complementary GUI here developed, but also as an individual package and by any external code via an Application Programming Interface (API), opening the possibility of integrating the code with third-party apps and web services. Last, the code is published open source code to facilitate collaborative, third-party development

of new functionalities. A complete documentation of the codebase can be found in Ref. [23].

PRISMA currently supports three main processing functionalities, summarized in Figure 1.

Spectrum trimming: Cut a spectrum at user-defined intervals.

Baseline correction: Fits a baseline curve to a spectrum based on the Asymmetric Least-Squares smoothing method (AsLS) developed by Eilers and Boelens.^[24,25] While there are multiple approaches for baseline correction of spectra,^[26] which could also readily be implemented as part of PRISMA's codebase, we prefer the AsLS method due to its relative simplicity, fast computation performance and excellent baseline estimation.^[25] Furthermore, unlike traditional approaches, the user does not need to manually indicate anchor points in the spectrum to estimate its baseline, thus minimizing user intervention and facilitating the automation of the correction process. The method finds a curve being i) smooth, ii) faithful to the spectrum, and iii) asymmetrically penalizing positive residuals, where the analytical peaks are found. The optimal baseline signal z_i is found by minimizing iteratively the penalized least-squares function [Eq. (1)]:

$$S = \sum_i w_i (y_i - z_i)^2 + \lambda \sum_i (\Delta^2 z_i)^2 \quad (1)$$

where y_i represents the spectrum counts and $\Delta^2 z_i = z_i - 2z_{i-1} + z_{i-2}$ quantifies the smoothness of the baseline. The user tunes the baseline through two main parameters. The first, the penalty p , accentuates the relative importance (the weights w_i) of data points at the base of the spectrum. Larger p include higher-valued data points in the fitting (Figure 2a). Second, the smoothness λ , controls the smoothness of the baseline curve. Larger λ result in flatter curves (Figure 2b).

Peak fitting: Models a spectrum as a combination of peak profiles defined by an analytical expression. For each peak, the user specifies an upper and a lower bound defining the peak neighbourhood, along with a maximum bound for the peak's width. The code then fits the complete spectrum as a sum of peak profiles using the Trust Region Reflective Algorithm as implemented in the SciPy package.^[27] The resulting fitting parameters (e.g., position, height and half width at half maximum) are used to characterize the spectrum. The peak profiles included in PRISMA (Table 1) describe the most commonly found peak profiles, which are naturally (Lorentzian),

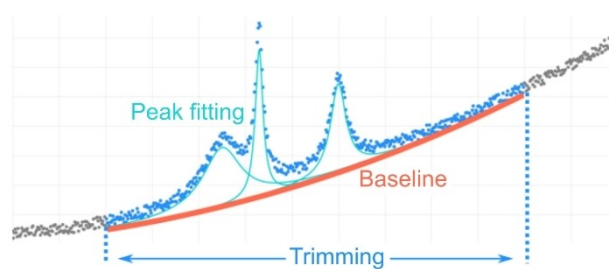


Figure 1. Representation of the three main functionalities currently implemented in PRISMA on a synthetic spectrum.

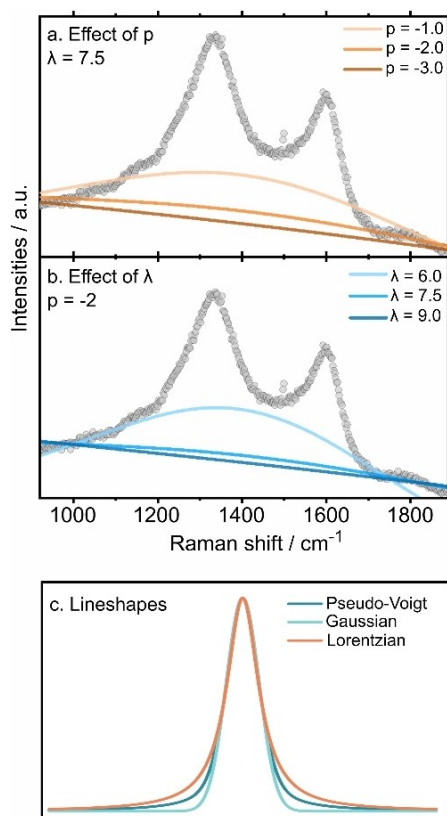


Figure 2. Effect of (a) penalty (p) and (b) smoothness (λ) parameters on the baseline of the Raman spectrum of hard carbon. (c) Peak profiles implemented in PRISMA.

Table 1. Three peak profiles implemented in PRISMA, defined by the peak height h , the half-width at half-maximum w and the centre p .

Peak Profile	Description	Analytical expression
Lorentzian	Bands where natural broadening is dominant. ^[28,30]	$\frac{hw^2}{(x-p)^2 + w^2}$
Gaussian	Bands dominated by measurement broadening.	$he^{-\frac{\ln(2)(x-p)^2}{w^2}}$
Pseudo-Voigt (50% Lorentzian)	Intermediate case between Lorentzian and Gaussian profiles.	$\frac{1}{2}L(x) + \frac{1}{2}G(x)$

partially (Pseudo-Voigt) or heavily (Gaussian) broadened by the measurement conditions.^[28,29] Figure 2c highlights differences among peak profiles: Lorentzian profiles exhibit longer tails compared to Gaussians, while Pseudo-Voigt profiles with 50% Gaussian/Lorentzian mix result in an intermediate profile.

The typical workflow when using PRISMA to analyse a sequence of dependent patterns is illustrated in the code documentation^[23] and the Supporting Information video. In short, the user:

- Chooses a pipeline, that is, a pre-defined recipe of spectrum processing steps.
- Uploads the file(s) with the patterns. PRISMA offers three parsers able to load patterns in specific formats; that is, as individual .txt files, as a single .txt file and as a single .csv

file. For more details on the accepted file formats see Supporting Information Table T1.

- Selects a spectrum to visualize.
- Tunes the processing parameters until reaching a visually satisfactory fit. For baseline correction, the parameters are the trimming interval and the λ and p parameters of the fit; for peak fitting these are the number of peak profiles, their bounding neighbourhood and a maximum width limit.
- Inspects whether the parameters enable satisfactory fits of other patterns or if they need further tuning.
- Iterates between steps d. and e. selecting multiple patterns and evaluating the results by visual inspection, until finding satisfactory parameters that apply to most patterns.
- Runs a high-throughput processing of all patterns with the chosen parameters.
- Downloads the results for further analysis and plotting. For more details on the output files see Supporting Information Table T2.

PRISMA also supports downloading individual .csv files of a processed spectrum that include baseline and peak profiles. This functionality renders the app suitable to analyse ex situ, post-mortem and, in general, patterns generated from unrelated experiments. In this case, the previous workflow includes an intermediate step to download the results of an individual spectrum and stops at step d. after all patterns have been processed and downloaded individually.

In the following sections we demonstrate three representative battery examples using PRISMA. The cases studies aim to test the user experience, robustness to different datasets and illustrate the scientific value of PRISMA in accelerating the processing of raw patterns.

Results and Discussion

Melting Transition of Ethylene Carbonate

Ethylene carbonate (EC) is an organic solvent widely used as component of Li-ion battery electrolytes due to its crucial role in forming stable solid electrolyte interphases (SEI).^[31] EC is solid at room temperature but melts at 36.3 °C.^[32] We have recorded the Raman spectra of EC as a function of temperature, from 25 °C to 65 °C in order to monitor the melting transition. Figure 3a shows a contour plot of the resulting temperature-dependent spectra within the interval 2825 to 3150 cm⁻¹ where CH₂ vibrations appear^[33] (complete range in Supporting Information Figure S1). Figure 3b shows samples of solid and liquid EC for comparison. Evidently, the spectra of solid and liquid phases remain stable except within the transition region between 35 and 40 °C, where most bands shift and change intensity.

The spectra within the 2825 to 3150 cm⁻¹ wavenumber region were processed with PRISMA for baseline correction and to obtain peak fitting parameters based on three Lorentzian spectral contributions, named for convenience ν_1 , ν_2 and ν_3 . Figure 4 shows the peak position and intensity trends as a function of temperature. As the transition temperature is

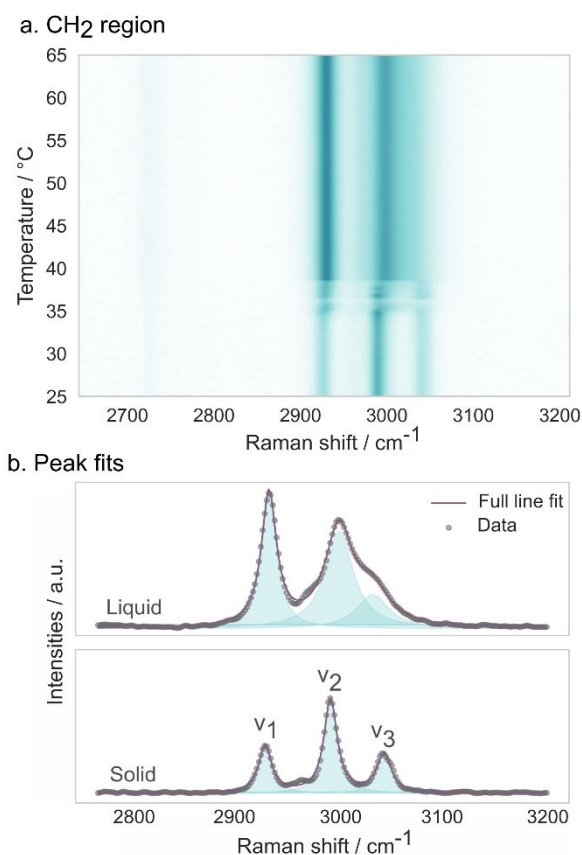


Figure 3. (a) Raman spectra of ethylene carbonate as a function of temperature in the range 2825 to 3150 cm^{-1} . (b) Raman spectrum and the PRISMA fits of liquid and solid ethylene carbonate.

approached, the bands shift in a non-monotonic fashion: ν_3 from 3040 to 3030 cm^{-1} , ν_2 from 2990 to 2998 cm^{-1} and ν_1 from 2926 to 2930 cm^{-1} . At the same time, the peak intensities of the two higher wavenumber C–H stretch bands vary very slightly, while the intensity of the lower wavenumber C–H stretching mode increases almost 3-fold during the state transition.

The spectroscopic changes likely originate from the structural rearrangement and concomitant symmetry changes the EC molecule undergoes upon melting.^[33–36] Notably, all trends display a clear discontinuity at the transition temperature.

PRISMA enables processing of the 100+ Raman spectra generated in the experiment within a few minutes. The analysis results in feature-rich peak parameters trends, which constitute an alternative and more detailed representation of the system's behaviour. For instance, the intensity discontinuities in our case study would be difficult to observe in conventional spectrum representations such as contour (e.g., Figure 3a) or waterfall plots, but they are clearly visible in the representation of Figure 4a–b. In a broader sense, the peak parameter representation enabled by tools like PRISMA highlights peak shifts, intensity changes and trend discontinuities; all of which can prove crucial in understanding the solid-liquid, liquid-gas, crystalline-amorphous transitions, etc., inherent to the operation and stability of a wide range of functional materials.

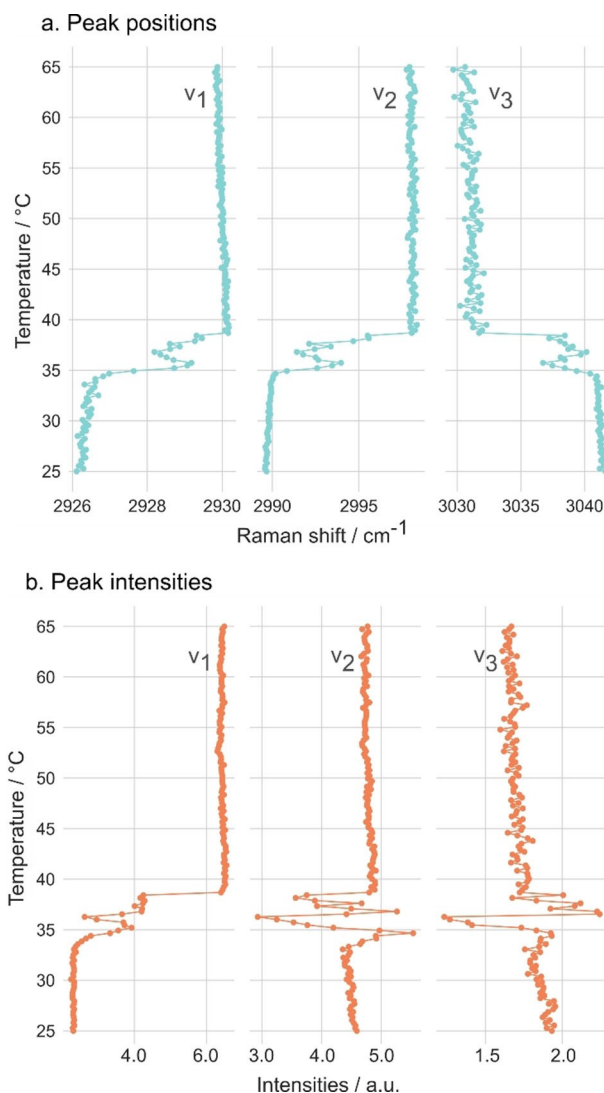


Figure 4. Peak parameters fitted with PRISMA: (a) positions and (b) intensities of the C–H stretch region of ethylene carbonate, as a function of temperature.

Raman Mapping of a Composite Graphite Electrode

Graphite is the anode material of choice for the Li-ion battery.^[37] Typically, the electrode is fabricated as a composite of micro-metre-sized graphite particles mixed with a conductive additive and coated with a binder onto a copper foil current collector (see image on the Figure 5d). Here, we performed line scan of the graphite electrode along 60 μm , recording a Raman spectrum every 0.5 μm , which oversamples the scanning line given that the lateral resolution is 2.4 μm . As the laser probes the electrode along the line shown in the Figure 5d, it encounters graphite particles fully or partially covered with the carbon conductive additive and binder, and particle interspaces, that is, voids to be filled with electrolyte. The subtle differences between the Raman signatures of graphite and the carbon additive introduce spectroscopic changes along the scanning direction.

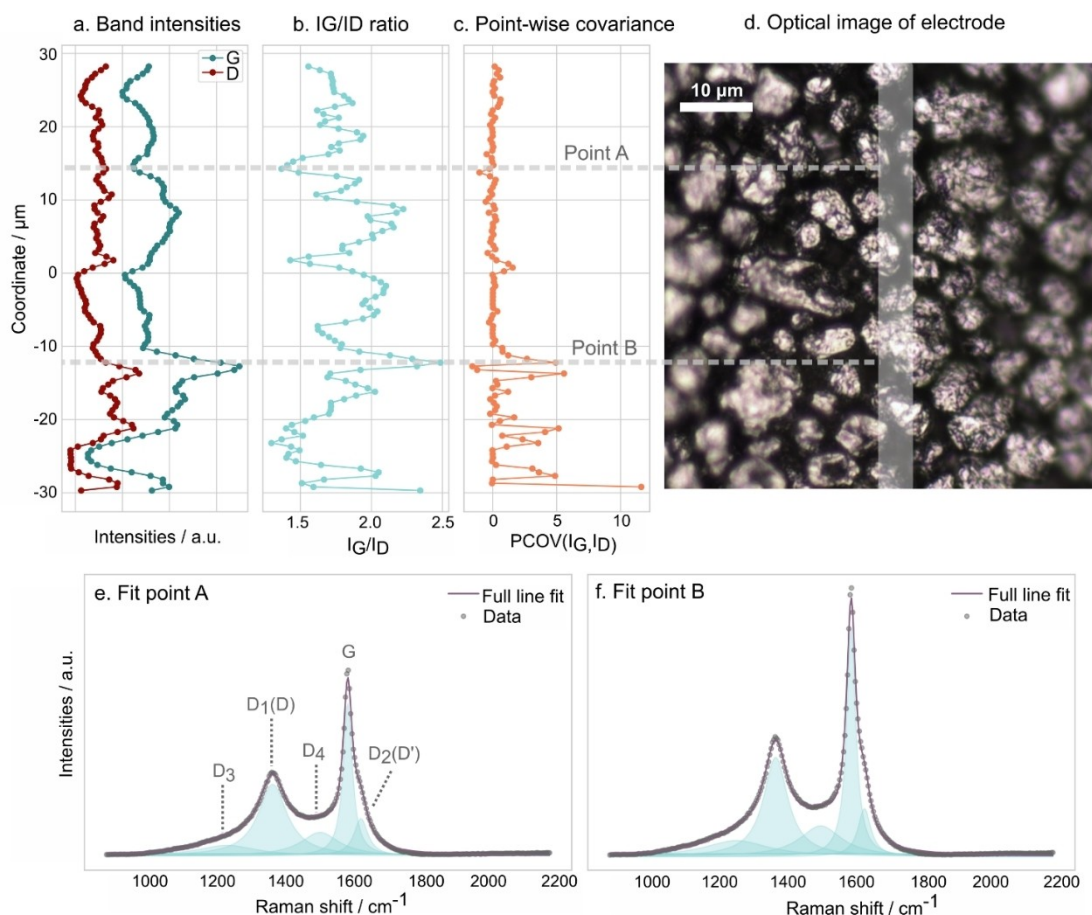


Figure 5. Peak fitting of the line scan case study. (a) Fitted D and G band intensities, (b) Intensity ratio trend, and (c) point-wise covariance between I_G and I_D numerical differences, as a function of the linear coordinate. (d) Aligned optical image of the electrode, highlighting the line scan. Baseline-subtracted spectra fitted with the 5-Lorentzian band model from (e) Point A and (f) Point B along the linear mapping.

Graphite, as well as other carbonaceous materials, is typically characterized by D and G Raman-active bands that appear within the 1000–1700 cm⁻¹ region and correspond to disordered and ordered domains in the graphitic structure, respectively.^[38] The ratio between the two bands depends on multiple variables, including incident laser wavelength, degree of disorder, defects, surface groups, and crystalline size.^[39] Disordered graphitic materials also exhibit a D' band located at a slightly higher wavenumber than the G band and typically appear as a shoulder of the latter. In some cases, three bands are not sufficient for an adequate fitting of spectra from a composite graphite electrode.^[40] A five-peak fitting model with D₁(D), D₂(D'), D₃, D₄ and G bands has been suggested to provide more accurate fits and insight into the structure and chemical nature of carbonaceous materials.^[41,42] The additional two bands, D₃ and D₄ have been assigned to presence of amorphous carbon from the conductive additive, surface groups and impurities.^[41,42] Figures 5e and 5f illustrate the 5-peak fitting model using Lorentzian peak profiles, from which the intensities are extracted and followed across the electrode.

Figure 5a shows the fitted D and G intensities along the line scan, aligned with the optical image in Figure 5d. Given there is no autofocus functionality enabled during the scan, the

recorded intensities are subjected to both i) topographic contrasts, originating from elevation changes relative to the axial position of the laser probe and ii) chemical contrasts, resulting from the mixture between graphitic and disordered carbon. We expect topographic changes (voids and hills) to affect the intensities of both D and G bands equally. Instead, chemical changes (order and disordered carbon) would predominantly affect the I_G to I_D intensity ratio.^[43] Based on these assumptions, we attempt deconvoluting topographic from chemical contrasts by computing two derived metrics: i) the I_G/I_D intensity ratio, and ii) a point-wise covariance between the numerical differences of I_G and I_D trends. The covariance metric was chosen because its property to increase when I_G and I_D change simultaneously and decrease when I_G and I_D changes are uncorrelated. In other words, high I_G/I_D ratios indicate graphite-rich regions, while high covariances are a proxy for elevation changes along the scanning direction.

The I_G/I_D trend in Figure 5b aids identifying highly crystalline graphitic regions; for instance, Point A coincides with a particle interspace that is filled with the disordered conductive carbon, according to the low I_G/I_D ratio. In contrast, the I_G/I_D ratio is high at Point B, where a big graphitic particle is clearly identified from the optical image. Interestingly, the covariance trends in

Figure 5c appear decoupled from the I_G/I_D ratios, suggesting a successful deconvolution of chemical and topographic contrasts. The covariance trends indicate a relatively flat region between 30 and $-10\ \mu\text{m}$ suggesting that particle interspaces are filled with conductive additive. Conversely, elevation changes appear between -10 and $-30\ \mu\text{m}$ and thus suggest the presence of large particles partially covered with conductive additive.

From the scientific perspective, this case study further illustrates the value of analysing peak trends instead of raw spectra. First, the baseline subtraction step removes the underlying intensity contributions from the baseline and enables exclusive monitoring of the Raman-active signals. Second, derivative metrics computed from the peak trends might offer insightful descriptions, for example, topographic contrasts, chemical heterogeneities, etc. From the user's perspective, the challenging spectroscopic analysis of carbonaceous materials illustrates another advantage of PRISMA: its GUI enables a fluent feedback between tuning the analysis and visualizing the results. The user is able to quickly explore many analysis constraints –baseline parameters, number and constrains of peaks– in the pursuit for a fit both visually accurate and scientifically sound.

Operando X-ray Diffraction of Li_xNiO_2

Challenges on the world supply of cobalt are fuelling research efforts on the search for stable, high-energy and Co-free cathode materials for Li-ion batteries.^[44,45] LiNiO_2 (LNO) is being revisited as a prominent candidate since it shares the layered crystal structure of already commercialized LiCoO_2 and NMC ($\text{LiNi}_x\text{Mn}_y\text{Co}_z\text{O}_2$) materials and deliver higher energy density; however, a series of cycling-dependent phase transitions are believed to limit its cycling reversibility.^[46] Therefore, it is important to characterize the phase diagram and the structural relation between the observed phases in order to identify the structural culprits of cell degradation. We have performed *operando* XRD experiments during delithiation and lithiation of a LNO/Li half-cell. The evolution of the XRD patterns from the Li_xNiO_2 electrode during galvanostatic cycling are characterized by multiple transitions among four phases (see Figure 6a): the initial hexagonal structure (H1) transforms into a monoclinic phase (M) during delithiation. By further delithiation another hexagonal phase (H2) is formed, which at high degree of delithiation transforms into an hexagonal phase (H3) with a significantly smaller interlayer distance. The reactions were shown to be reversible upon subsequent lithiation.^[47,48]

The phase transitions of Li_xNiO_2 upon delithiation are accompanied with changes of lattice parameters and relative fraction of distinct phases, which will reflect on diffraction peak positions and intensities, respectively. These changes originate from a combined effect of the lithium-content (x) and Ni^{3+} to Ni^{4+} oxidation during Li_xNiO_2 delithiation.^[49,50] Particularly, the c lattice parameter is known to undergo a dramatic decrease when H2 phase transforms to H3 because the Li-containing intralayer collapses at low lithium-content (x).^[51] The large

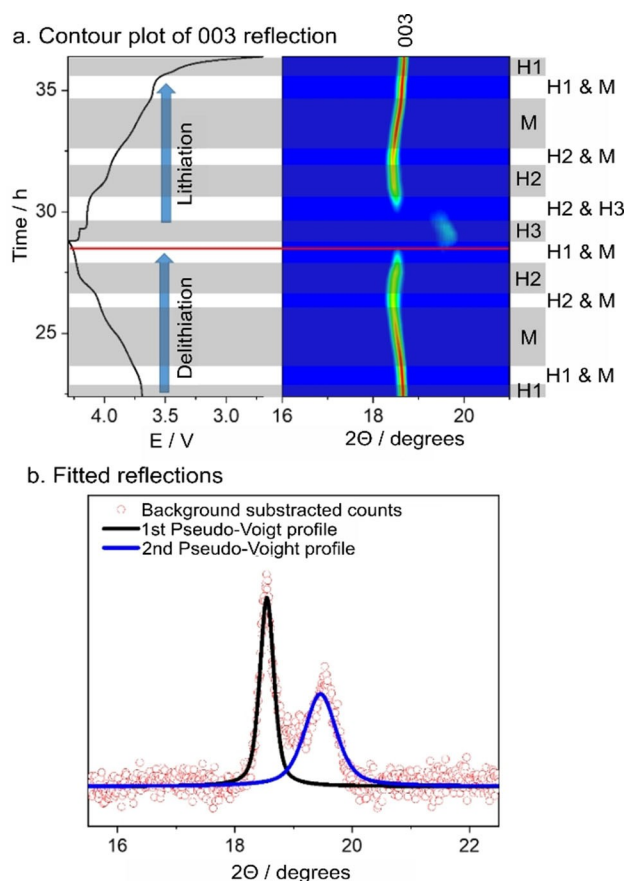


Figure 6. (a) XRD contour plot of the 003 reflection of Li_xNiO_2 and voltage profile as a function of time during delithiation and lithiation. Single-phase regions are indicated using grey rectangle shade and biphasic regions are indicated between these single-phase regions; (b) example of PRISMA-fitted XRD pattern within H2 and H3 biphasic region (Red line indicates the selected XRD pattern).

difference in lattice parameter c between H2 and H3 phases manifests as a shift of the 003 reflection of Li_xNiO_2 in XRD patterns.

We performed a HT fitting of the 003 reflection using PRISMA to obtain background-subtracted XRD patterns and peak fitting parameters including peak positions (d -spacing) and heights. The baseline subtraction procedure enabled decoupling the effects of interfering signals from the glassy carbon used as cell window (see Supporting Information Figure S2). Two pseudo-Voigt peak profiles were adapted to fit the 003 reflection: the first fits the 003 reflection at low angle ($18.4^\circ \leq 2\theta \leq 18.7^\circ$), corresponding to H1/M/H2 phase regions; and the second fits the H3 phase at high angle ($19.5^\circ \leq 2\theta \leq 19.7^\circ$). Figure 6b shows an example of a PRISMA-fitted XRD pattern within the H2/H3 biphasic region. Evolutions of d -spacings and peak heights for the 003 reflection as a function of time upon delithiation and lithiation are presented in Figure 7a and b, respectively. The obtained evolution of d -spacing is highly consistent with that from Rietveld refinements and clearly shows a significant d_{003} change between H1 and H3 phases. In addition, the peak intensity trends show the H3 phase appearing at cell voltages larger than 4.1 V versus Li^+/Li

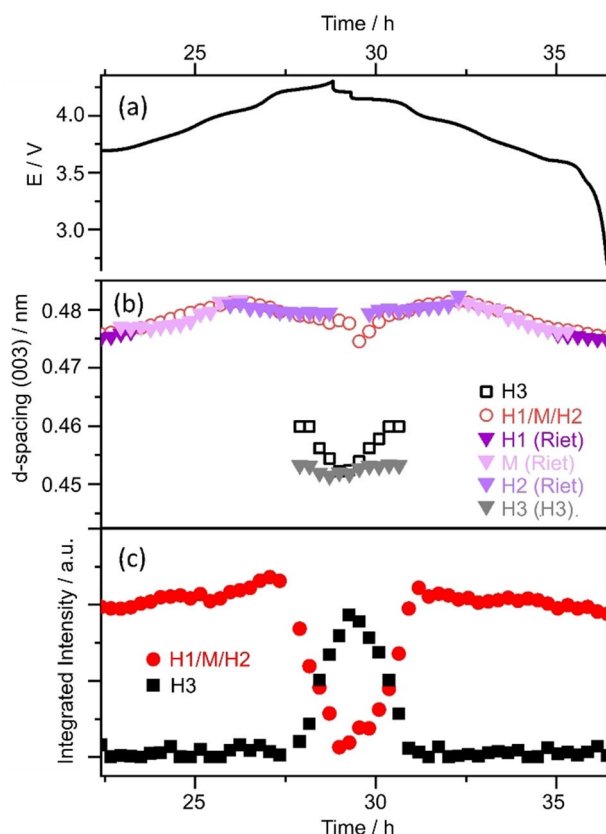


Figure 7. (a) Voltage profile as a function of time during delithiation and lithiation; (b) Comparison of obtained d-spacing 003 evolution upon delithiation and lithiation from Rietveld refinement and PRISMA fitting; (c) PRISMA fitted peak heights of the 003 reflection upon delithiation and lithiation.

and becoming the dominant phase thereafter. The presence of a contracted H3 phase has been associated with intra- and inter-particle mechanical stresses leading eventually to particle cracking and the rapid degradation of the active material.^[48,52–54]

PRISMA is by no means a replacement for structural refinement for phase identification. However, when analysing hundreds of diffraction patterns, individual refinement becomes cumbersome and time consuming. Sequential refinement might automate the analysis of large number of patterns, but choosing the right initial structural model can be challenging for multi-phasic systems such as LNO. Here PRISMA becomes a complementary tool for quickly analysing many patterns within a narrow region of interest, for example the 003 reflection for LNO reflecting cycling-dependent changes on the d_{003} spacing. Moreover, using PRISMA as an element in automated structural analysis of powder diffraction data from in situ and operando experiments could enable autonomous experimental control by providing on-line and real-time interpretation of crystallographic properties.

Conclusion

We introduce PRISMA as a general purpose and user-friendly tool for high-throughput analysis of patterns from spectroscopy and diffraction measurements. The app abstracts away implementation details and enables researchers to focus on applying their knowledge and intuition for analysing hundreds of patterns. Users execute an intuitive workflow essentially loading the raw files, tuning analysis parameters, visually inspecting the results and running a high-throughput processing step – all executed over few minutes. The graphical user interface is intuitive and provides a flexible control over the analysis parameters, which are saved for guaranteeing the reproducibility of the results. The app accepts general spectroscopic files (including diffraction patterns) with widely used text formats, and outputs the results as .csv files that can be easily loaded to most scientific plotting software. We test the current version of PRISMA with three experimental cases frequently encountered in electrochemical energy storage research. The cases demonstrate i) the robustness of PRISMA to analyse spectroscopic and diffraction data with varying degrees of noise and baselines and ii) the benefits of analysing peak trends for understanding the physicochemical phenomena underpinning the operation of functional materials.

More broadly, PRISMA also aims to become an interoperable platform connecting the development of pattern-processing tools, to their implementation, prototyping, and final deployment for general users. In this sense, we envision adding new tools i) to complement the current analysis pipelines (e.g., with outlier rejection and smoothing), or ii) to automate processing of patterns with fully autonomous algorithms,^[7,26,55,56] or iii) to implement data-driven analysis via dimensionality reduction techniques^[57,58] and trained machine learning models.^[59]

Experimental Section

Case of melting transition of ethylene carbonate. Temperature-dependent Raman measurements were carried out on 1 ml of ethylene carbonate (E-lyte) in a glass vial placed inside a custom-built temperature control chamber. The temperature ramp was set to heat from 25 °C to 65 °C within 7 h, using a digital temperature controller Eurotherm. The Raman spectra were acquired with a Bruker MultiRAM FT-Raman spectrometer fitted with a liquid nitrogen cooled Ge-diode detector, probing with a 1064 nm laser wavelength at 400 mW of laser power. A total of 150 spectra were recorded over 100 scans at 4 cm⁻¹ resolution, each at 0.26 °C temperature intervals.

Case of Raman mapping of a composite graphite electrode. The composite graphite electrode was manufactured by CIDITEC using the following composition: 15–4 Graphite (IMERYS), 94 wt%; C45 Carbon Black (IMERYS), 2 wt%; BM451B SBR (ZEON), 2 wt%; Carboxymethyl cellulose binder: Walocel CRT2000(DOW), 2 wt%. The Raman spectra were acquired with a HORIBA LabRam HR Evolution confocal Raman spectrometer using 50x long working distance objective and 488 nm laser wavelength (filtered to 5% laser power, c.a. 2.5 mW), 300 nm grating and 150 seconds total acquisition time. The confocal hole was set to 200 nm. The line scan was recorded over 60 μm with a spectrum recorded every 0.5 μm.

Case of Operando X-ray diffraction of LiNiO₂. The LiNiO₂ slurry was prepared by mixing LiNiO₂ active material powder (BASF), PVDF (Sigma Aldrich) and carbon black (IMERYS, ≥96%) with a weight ratio of 93: 3.5: 3.5 in N-Methyl-2-pyrrolidone solvent. The slurry was coated on an aluminium foil and dried at 120 °C overnight. The dried material was scratched from the foil and cold-pressed into a pellet (7 mm in diameter and 152 µm in thickness) under 0.75 tons pressure and subsequently dried at 120 °C in a vacuum oven inside a glovebox overnight. A modified electrochemical-cell (ECC-OPTO) was assembled inside the glovebox using the pellet as cathode, a lithium foil with a diameter of 9 mm as anode and LP57 (E-lyte, 1 M LiPF₆ in 3:7 EC/EMC) as electrolyte for operando X-ray diffraction (XRD) measurements; further details about the cell and its assembly can be found in Ref. [60]. Diffraction data were recorded on a Rigaku Smartlab diffractometer in reflection geometry with a Cu_{Kα} ($\lambda_1 = 1.54059$, $\lambda_2 = 1.54441$ Å; λ_2/λ_1 intensity ratio=0.5) radiation source. The diffraction data were collected every 16.5 minutes within a 2θ range of 10–130° with a step size of 0.01°.

CRediT authorship contribution statement

Eibar Flores: Conceptualization, Software development, Formal Analysis, Investigation, Visualization, Writing – original draft, Writing – Review & editing, Project Administration. Nataliaia Mozkhukhina: Resources, Experiments, Investigation, Writing – original draft for cases 1 and 2. Xinyu Li: Resources, Experiments, Investigation, Writing – Writing – original draft for case 3. Poul Norby: Supervision, Writing – Review & editing for case 3. Aleksandar Matic: Supervision, Writing – Review & editing for cases 1 and 2. Tejs Vegge: Funding acquisition, Conceptualization, Writing – Review & editing.

Supporting Information Summary

Supporting Information with Table of supported file-formats, contour Raman plots of ethylene carbonate and contour Raman plots of original and baseline-corrected XRD pattern of LiNiO₂. Supporting Video tutorial for using the software.

Acknowledgements

This project has received funding from the European Union's Horizon 2020 research and innovation initiative under grants agreement No 957189 (BIG-MAP) and No 957213 (BATTERY2030-PLUS).

Conflict of Interest

The authors declare no conflict of interest.

Data Availability Statement

Research data are not shared.

Keywords: electrochemical energy storage · high throughput analysis · in situ spectroscopy · operando spectroscopy · software

- [1] J. Lu, T. Wu, K. Amine, *Nat. Energy* **2017**, 2, 17011.
- [2] Z. Lin, P. L. Taberna, P. Simon, *Curr. Opin. Electrochem.* **2018**, 9, 18–25.
- [3] A. Arvay, E. Yli-Rantala, C. H. Liu, X. H. Peng, P. Koski, L. Cindrella, P. Kauranen, P. M. Wilde, A. M. Kannan, *J. Power Sources* **2012**, 213, 317–337.
- [4] X. Li, H.-Y. Wang, H. Yang, W. Cai, S. Liu, B. Liu, *Small Methods* **2018**, 2, 1700395.
- [5] P. P. R. M. L. Harks, F. M. Mulder, P. H. L. Notten, *J. Power Sources* **2015**, 288, 92–105.
- [6] PeakUtils: Lucas Hermann Negri, **2017**, DOI 10.5281/zenodo.887917.
- [7] T. M. James, M. Schlösser, R. J. Lewis, S. Fischer, B. Bornschein, H. H. Telle, *Appl. Spectrosc.* **2013**, 67, 949–959.
- [8] P. Candeloro, E. Grande, R. Raimondo, D. D. Mascolo, F. Gentile, M. L. Coluccio, G. Perozziello, N. Malara, M. Francardi, E. Di Fabrizio, *Analyst* **2013**, 138, 7331–7340.
- [9] F. Menges, **2017**, “Spectragryph-optical spectroscopy software”, can be found under <https://effemm2.de/spectragryph/index.html>.
- [10] E. Flores, N. Vonrüti, P. Novák, U. Aschauer, E. J. Berg, *Chem. Mater.* **2018**, 30, 4694–4703.
- [11] E. Flores, P. Novák, U. Aschauer, E. J. Berg, *Chem. Mater.* **2020**, 32, 186–194.
- [12] E. Flores, N. Mozkhukhina, U. Aschauer, E. J. Berg, *ACS Appl. Mater. Interfaces* **2021**, 13, 22540–22548.
- [13] N. Mozkhukhina, E. Flores, R. Lundström, V. Nyström, P. G. Kitz, K. Edström, E. J. Berg, *J. Phys. Chem. Lett.* **2020**, 11, 4119–4123.
- [14] T. Vegge, J.-M. Tarascon, K. Edström, *Adv. Energy Mater.* **2021**, 11, 2100362.
- [15] M. Fichtner, K. Edström, E. Ayerbe, M. Bercebar, A. Bhowmik, I. E. Castelli, S. Clark, R. Dominko, M. Eracka, A. A. Franco, A. Grimaud, B. Horstmann, A. Latz, H. Lorrman, M. Meeus, R. Narayan, F. Pammer, J. Ruhland, H. Stein, T. Vegge, M. Weil, *Adv. Energy Mater.* **2021**, 2102904.
- [16] I. E. Castelli, D. J. Arismendi-Arrieta, A. Bhowmik, I. Cekic-Laskovic, S. Clark, R. Dominko, E. Flores, J. Flowers, K. Ulvskov Frederiksen, J. Friis, A. Grimaud, K. Vels Hansen, L. J. Hardwick, K. Hermansson, L. Königer, H. Lauritzen, F. Le Cras, H. Li, S. Lyonard, H. Lorrman, N. Marzari, L. Niedzicki, G. Pizzi, F. Rahmanian, H. Stein, M. Uhrin, W. Wenzel, M. Winter, C. Wölke, T. Vegge, *Batteries & Supercaps* **2021**, batt.202100117.
- [17] C. R. Harris, K. J. Millman, S. J. van der Walt, R. Gommers, P. Virtanen, D. Cournapeau, E. Wieser, J. Taylor, S. Berg, N. J. Smith, R. Kern, M. Picus, S. Hoyer, M. H. van Kerkwijk, M. Brett, A. Haldane, J. F. del Río, M. Wiebe, P. Peterson, P. Gérard-Marchant, K. Sheppard, T. Reddy, W. Weckesser, H. Abbasi, C. Gohlke, T. E. Oliphant, *Nature* **2020**, 585, 357–362.
- [18] “ipywidgets,” can be found under <https://github.com/jupyter-widgets/ipywidgets>, **2021**.
- [19] T. Kluyver, B. Ragan-Kelley, F. Pérez, B. Granger, M. Bussonnier, J. Frederic, K. Kelley, J. Hamrick, J. Grout, S. Corlay, P. Ivanov, D. Avila, S. Abdalla, C. Willing, *Position. Power Acad. Publ. Play. Agents Agendas - Proc. 20th Int. Conf. Electron. Publ. ELPUB* **2016**, 87–90.
- [20] “Voila,” can be found under <https://github.com/voila-dashboards/voila>, **2021**.
- [21] H. Happel, S. Seedorf, in *Work. Semat. Web Enabled Softw. Eng. ISWC*, **2006**, pp. 5–9.
- [22] S. P. Huber, S. Zoupanos, M. Uhrin, L. Talirz, L. Kahle, R. Häuselmann, D. Gresch, T. Müller, A. V. Yakutovich, C. W. Andersen, F. F. Ramirez, C. S. Adorf, F. Gargiulo, S. Kumbhar, E. Passaro, C. Johnston, A. Merkys, A. Cepellotti, N. Mounet, N. Marzari, B. Kozinsky, G. Pizzi, *Sci. Data* **2020**, 7, 300.
- [23] E. Flores, “PRISMA” can be found under <https://github.com/BIG-MAP/PRISMA>, **2021**.
- [24] P. H. C. Eilers, *Anal. Chem.* **2003**, 75, 3631–3636.
- [25] P. H. C. Eilers, H. F. M. Boelens, *Baseline Correction with Asymmetric Least Squares Smoothing*, **2005**.
- [26] G. Schulze, A. Jirasek, M. M. L. Yu, A. Lim, R. F. B. Turner, M. W. Blades, *Appl. Spectrosc.* **2005**, 59, 545–574.
- [27] P. Virtanen, R. Gommers, T. E. Oliphant, M. Haberland, T. Reddy, D. Cournapeau, E. Burovski, P. Peterson, W. Weckesser, J. Bright, S. J. van der Walt, M. Brett, J. Wilson, K. J. Millman, N. Mayorov, A. R. J. Nelson, E. Jones, R. Kern, E. Larson, C. J. Carey, Í. Polat, Y. Feng, E. W. Moore, J. VanderPlas, D. Laxalde, J. Perktold, R. Cimman, I. Henriksen,

- E. A. Quintero, C. R. Harris, A. M. Archibald, A. H. Ribeiro, F. Pedregosa, P. van Mulbregt, *Nat. Methods* **2020**, *17*, 261–272.
- [28] W. Demtröder, in *Laser Spectroscopy*, Springer, Berlin, **2003**, pp. 59–95.
- [29] T. Váczi, *Appl. Spectrosc.* **2014**, *68*, 1274–1278.
- [30] R. J. Meier, *Vib. Spectrosc.* **2005**, *39*, 266–269.
- [31] K. Xu, *Chem. Rev.* **2004**, *104*, 4303–4418.
- [32] M. S. Ding, K. Xu, T. R. Jow, *J. Electrochem. Soc.* **2000**, *147*, 1688.
- [33] M. Masia, M. Probst, R. Rey, *J. Phys. Chem. B* **2004**, *108*, 2016–2027.
- [34] B. Fortunato, P. Mirone, G. Fini, *Spectrochim. Acta Part A* **1971**, *27*, 1917–1927.
- [35] J. L. Allen, O. Borodin, D. M. Seo, W. A. Henderson, *J. Power Sources* **2014**, *267*, 821–830.
- [36] B. Klassen, R. Aroca, M. Nazri, G. A. Nazri, *J. Phys. Chem. B* **1998**, *102*, 4795–4801.
- [37] J. Asenbauer, T. Eisenmann, M. Kuenzel, A. Kazzazi, Z. Chen, D. Bresser, *Sustain. Energy Fuels* **2020**, *4*, 5387–5416.
- [38] M. A. Pimenta, G. Dresselhaus, M. S. Dresselhaus, L. G. Cançado, A. Jorio, R. Saito, *Phys. Chem. Chem. Phys.* **2007**, *9*, 1276–1290.
- [39] A. C. Ferrari, J. Robertson, *Phys. Rev. B* **2000**, *61*, 14095–14107.
- [40] C. Mao, M. Wood, L. David, S. J. An, Y. Sheng, Z. Du, H. M. Meyer, R. E. Ruther, D. L. Wood, *J. Electrochem. Soc.* **2018**, *165*, A1837–A1845.
- [41] A. Sadezky, H. Muckenhuber, H. Grothe, R. Niessner, U. Pöschl, *Carbon* **2005**, *43*, 1731–1742.
- [42] X. Dou, I. Hasa, D. Saurel, C. Vaalma, L. Wu, D. Buchholz, D. Bresser, S. Komaba, S. Passerini, *Mater. Today* **2019**, *23*, 87–104.
- [43] M. A. Pimenta, G. Dresselhaus, M. S. Dresselhaus, L. G. Cançado, A. Jorio, R. Saito, *Phys. Chem. Chem. Phys.* **2007**, *9*, 1276–1291.
- [44] E. A. Olivetti, G. Ceder, G. G. Gaustad, X. Fu, *Joule* **2017**, *1*, 229–243.
- [45] Y.-K. Sun, D.-J. Lee, Y. J. Lee, Z. Chen, S.-T. Myung, *ACS Appl. Mater. Interfaces* **2013**, *5*, 11434–11440.
- [46] Y. Kobayashi, M. Tabuchi, H. Miyashiro, N. Kuriyama, *J. Power Sources* **2017**, *364*, 156–162.
- [47] W. Li, J. N. Reimers, J. R. Dahn, *Solid State Ionics* **1993**, *67*, 123–130.
- [48] H. Li, N. Zhang, J. Li, J. R. Dahn, *J. Electrochem. Soc.* **2018**, *165*, A2985–A2993.
- [49] T. Ohzuku, A. Ueda, M. Nagayama, *J. Electrochem. Soc.* **1993**, *140*, 1862–1870.
- [50] C. Delmas, J. P. Pérès, A. Rougier, A. Demourgues, F. Weill, A. Chadwick, M. Broussely, F. Perton, P. Biensan, P. Willmann, *J. Power Sources* **1997**, *68*, 120–125.
- [51] L. Croguennec, C. Poullierie, A. N. Mansour, C. Delmas, *J. Mater. Chem.* **2001**, *11*, 131–141.
- [52] L. De Biasi, A. O. Kondrakov, H. Geßwein, T. Brezesinski, P. Hartmann, J. Janek, *J. Phys. Chem. C* **2017**, *121*, 26163–26171.
- [53] W. S. Yoon, K. Y. Chung, J. McBreen, X. Q. Yang, *Electrochem. Commun.* **2006**, *8*, 1257–1262.
- [54] C. S. Yoon, D. W. Jun, S. T. Myung, Y. K. Sun, *ACS Energy Lett.* **2017**, *2*, 1150–1155.
- [55] V. I. Korepanov, *J. Raman Spectrosc.* **2020**, *51*, 2061–2065.
- [56] M. N. Schmidt, T. S. Alstrøm, **2019**, 2757–2761.
- [57] J. Felten, H. Hall, J. Jaumot, R. Tauler, A. De Juan, A. Gorzsás, *Nat. Protoc.* **2015**, *10*, 217–240.
- [58] V. P. Pauca, J. Piper, R. J. Plemmons, *Linear Algebra Appl.* **2006**, *416*, 29–47.
- [59] C. A. Meza Ramirez, M. Greenop, L. Ashton, I. ur Rehman, *Appl. Spectrosc. Rev.* **2020**, *0*, 1–31.
- [60] D. D. Blanchard, M. Slagter, *J. Phys. E* **2021**, *18*, 119–123.

Manuscript received: November 12, 2021
Version of record online: February 19, 2022